

## *Chapter 3 Properties and Characterization of Nanoparticles*

### 3.1 General properties of Nanoparticles

#### General

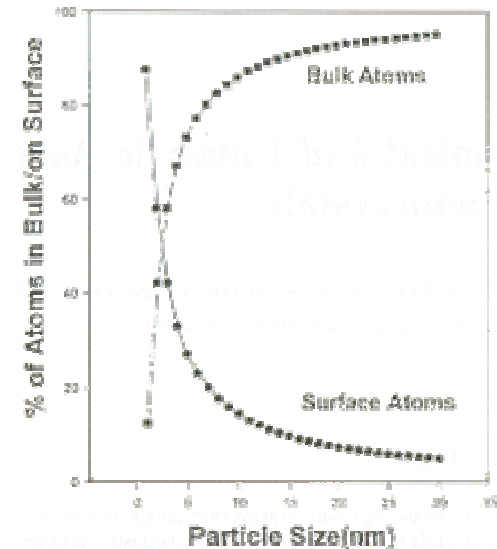
##### *\*Finite size effect*

- Small number of atoms
- quantum confinement (size) effect  
“particle-in-a box”
  - discrete electron energy levels
- perfect crystalline

##### *\*Surface/interface effect*

- Large fraction of surface atoms
- symmetry breaking at surface
  - large surface energy
  - changes in bond structure, atom coordination and lattice

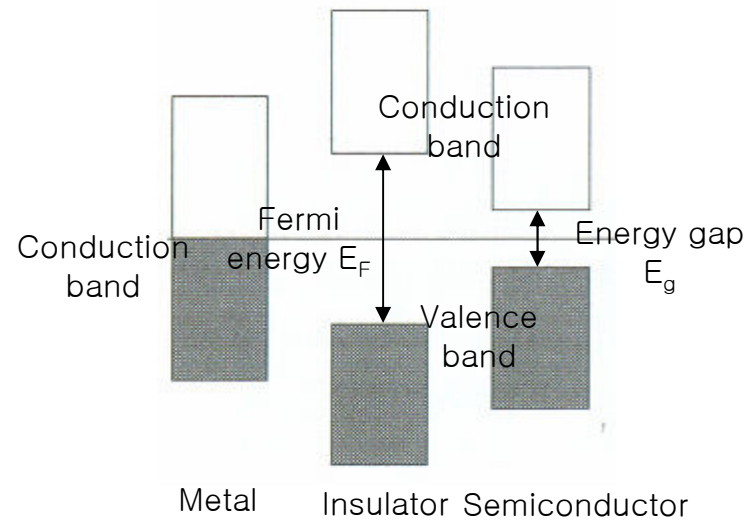
*constant*



Calculated values form spherical iron nanoparticles

### 3.2 Quantum size (confinement) effects

- Energy levels of atoms vs. energy band in bulk
- Conductor, semiconductor, insulator



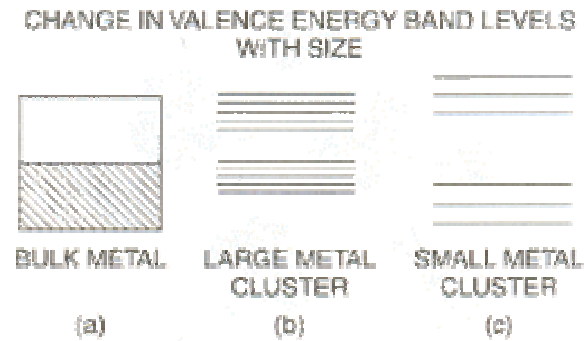
*As size decreases (< de Broglie wavelength\*)*

*electrons (and holes) are confined  $\rightarrow$  “particle in a box”*

*$\rightarrow$  electrons: discrete energy level or level spacing  $> k_B T$*

$\therefore$

## Metals



- *Energy level spacing for metal*

$$\delta = \frac{4E_F}{3N}$$

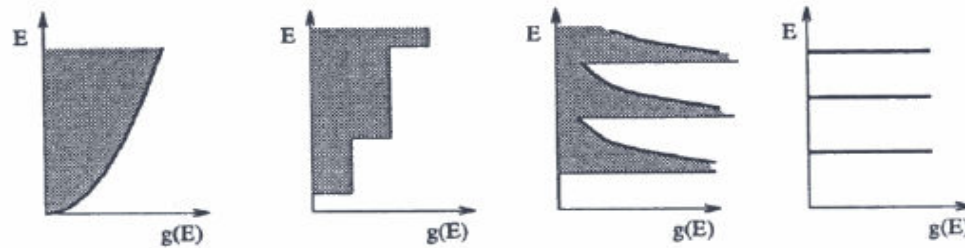
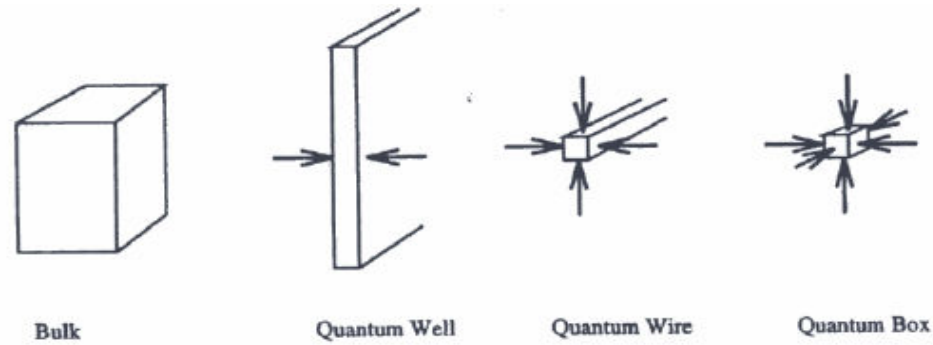
where  $E_F$ : Fermi energy

$N$ : number of electrons in the particles

- *level spacing: relatively very small*

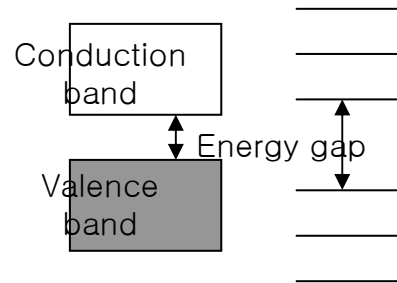
- *quantum size effect on electronic and optical properties: not significant even when  $N = \text{tens, hundreds...}$*

*\* Energy states of electrons for bulk, quantum well, wire and dots*



Change in the density of states,  $g(E)$ , as the number of confining dimensions is increased.

## Semiconductors



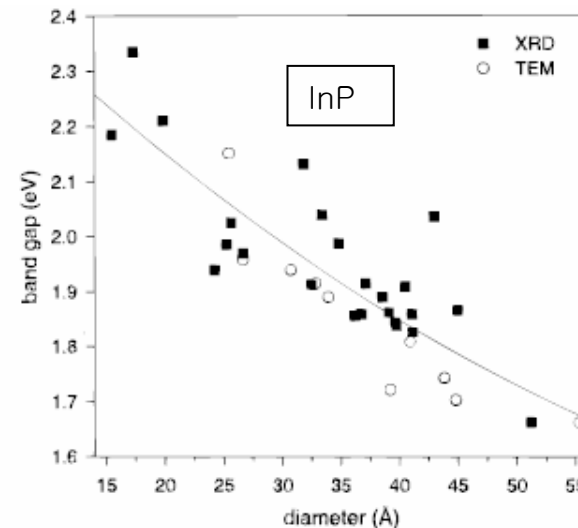
\* *Electron-hole pairs (excitons)*

- *Due to much longer wavelength of excitons in semiconductor (1 $\mu$ m compared to 0.5nm for metal) size confinement appears for  $N=10,000$  atoms*

- *Discrete energy levels*

- *Increase in band gap*

*cf. Insulators?*



Excitonic peak position from UV/vis vs nanocrystal diameter as determined by powder TEM (circles) or XRD (squares).

### 3.3 Structural effect

- Mostly related with surface properties...

*Au*

- <5nm: icosahedral rather than FCC (bulk)

*Al*

-  $Al_{13}$ : Icosahedral rather than FCC (bulk)

decrease in binding energy to 2.77eV from 3.39eV

decrease in Al separation to 2.814Å from 2.86Å

*In*

- <6.5nm: face-centered cubic rather than face-centered tetrahedral (>6.5nm)

( $c/a=1$ )

( $c/a=1.075$ )

*CdS*

- Lattice constants decreases linearly with  $1/d_p$

- Surface modification: less reduction in lattice constant

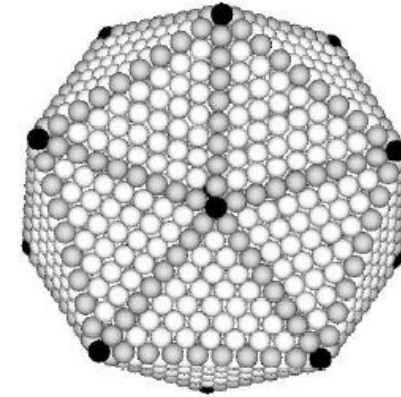


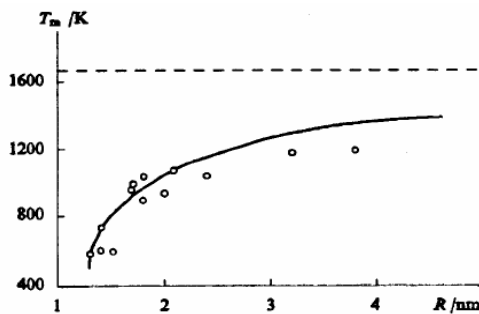
FIG. 1: An ideal Ih structure with the magic number of 2869 atoms. Atoms are shaded to indicate their local structure: fcc is white, hcp is gray, and other is black.

### 3.4 Thermodynamic effect

- due to large fraction of surface atoms

#### Lowering of melting point

Thermodynamic analysis



Melting point of CdS particles

$$T_b - T_m = \left[ \frac{4T_b}{\Delta H \rho_p d_p} \right] \left[ \sigma_p - \sigma_l \left( \frac{\rho_p}{\rho_l} \right)^{2/3} \right]$$

where  $T_b$ ,  $T_m$ : melting points of bulk and particle, respectively

$\Delta H$ : molar heat of fusion

-Similarly lowers other phase transition temperatures

- lowers sintering temperatures...

Melting point and sintering temperature for nanoparticles

	Nanoparticles	Bulk
Melting point	Au(3nm):900K In(4nm):370K	1300K 430K
Sintering temperature	Ni(20nm):~200°C W(22nm):~1100°C	>700°C >2000°C



Kelvin effect

- Vapor pressure

- Solubility

*Ostwald ripening*

### ***3.5 Chemical (catalytic) properties***

- *Surface activity*

*Bulk < plane surface < edge < corner*

- *Catalysts: mainly depends on metal nanoparticles...*
- *Supported on alumina, silica or charcoal (highly porous materials)*
  - *Usually made by impregnation of metal salt on support surface*
- *Followed by reduction*
- *Monodisperse metal nanoparticles: research issue...*
  - *Ligand protected (passivated) clusters*
  - *Reducing surface activity but tuning the activity...*

*Ex. 0.5wt% Pd NP modified with different ligands for reaction 25°C*

*2-hexene + H<sub>2</sub> = cis-2 hexene*

*with 1,10-phenanthroline : ultimately main product: trans-2 hexene*

*with 2-nbutylphenanthroline: slow reaction to 100% cis-form*

### 3.6 Mechanical properties

[http://www.wtec.org/loyola/nano/06\\_02.htm](http://www.wtec.org/loyola/nano/06_02.htm)

- Elastic (Young's) modulus ( $E$ ), yield strength, hardness, brittleness (or ductility)

\* Failure mechanism of conventional grain-sized materials

- Crack propagation
- Brittle-to-ductile transition due to dislocation

\* Nanocrystalline materials:

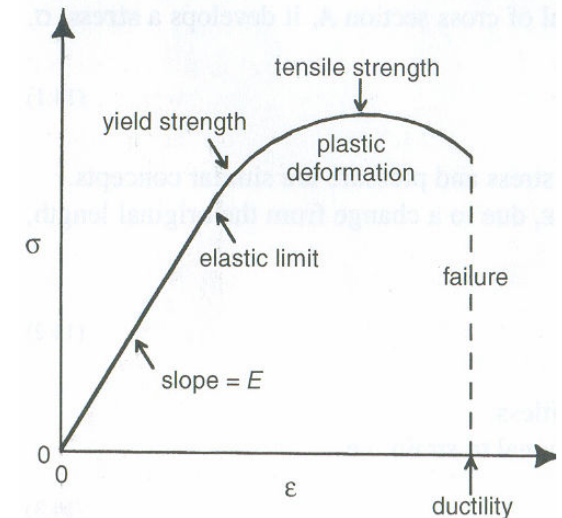
- dislocation-free or perfectness as grain diameter  $d$  decreases from 50-100nm.
- Some pores or flaw originated from preparation method

\* Whisker: perfectness appears for diameter  $< 10\mu\text{m}$ !

- full theoretical mechanical strength!

\* Deformation and fracture mechanism of NC materials: unclear

cf. for coarse grained materials: formation and movement of dislocation

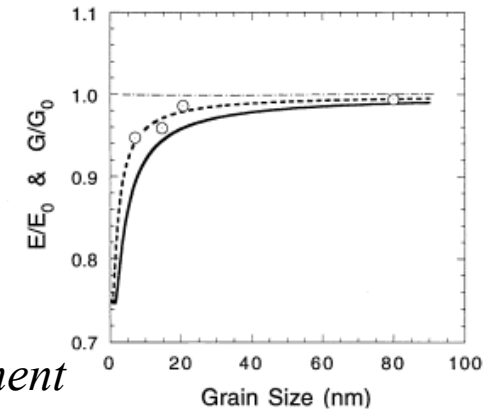


### Elastic properties

- For grain size  $< 5\text{nm}$ : decreasing  $E$  due to number of atoms associated grain boundaries becomes very large

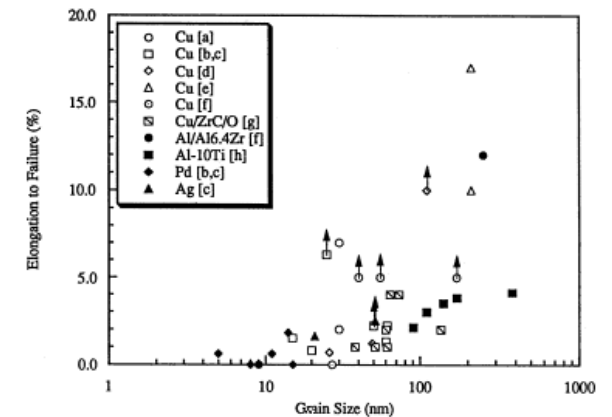
### Hardness and strength

- Increase with (grain diameter,  $d$ )<sup>-1/2</sup> (Hall-Petch equation)  
since grain boundaries: barrier against dislocation movement
- Some NC materials have showed deviation from H-P equation...



### Ductility and toughness

- Increase in coarse grain size materials ( $d > 1\mu\text{m}$ ),  
toughness and ductility increases
- NC materials ( $d < 30\text{nm}$ ): brittle



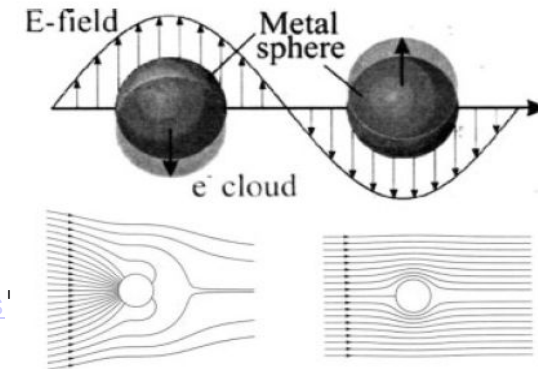
### Superplastic behavior

- : elongation (tensile deformation) of 100% to  $> 1000\%$  without necking or fracture
- Some NC: superplasticity transition temperature is lowered and strain rate increased with the decrease in  $d$ , usually for  $T > 0.5T_m$

### 3.7 Optical properties

- Due to surface plasmon resonance
- Due to the increased energy level spacing

[http://www.materialstoday.com/pdfs\\_7\\_2/marzan.pdf#search='nanoparticles'](http://www.materialstoday.com/pdfs_7_2/marzan.pdf#search='nanoparticles')



#### Surface plasmon resonance

: Coherent excitation of all the free electrons by light, leading to an in-phase oscillation for particles(  $d_p < \lambda_{light}$  )

→ Intense SP absorption bands

→ Enhancement of local electromagnetic field (surface enhancement resonance)

- SP energy determined by the dielectric property of the surrounding medium, the free electron density (size) of the particle.

- width of resonance (absorption band) related with surface scattering of electrons

- For small nanoparticles( $d_p \ll \lambda_{light}$ )

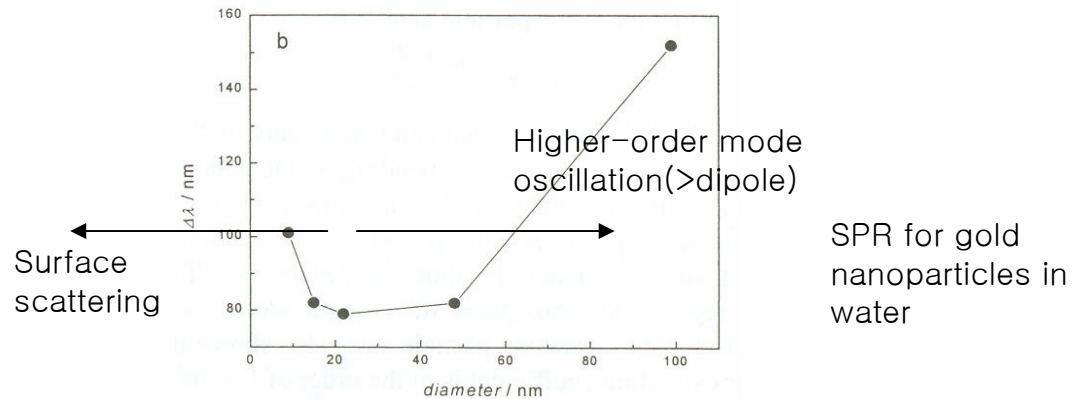
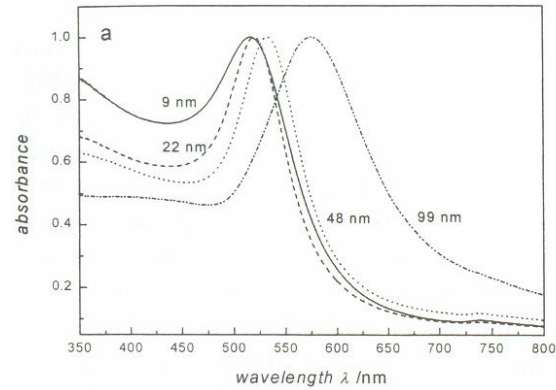
$$\sigma_{ext} = \frac{9\omega_m^{3/2} V \epsilon_2(\omega)}{c \{ [\epsilon_1(\omega) + 2\epsilon_m]^2 + \epsilon_2(\omega)^2 \}}$$

$\omega$  : angular frequency

$V$  : particle volume

$\epsilon_m, \epsilon$  : dielectric constants of medium and particle, where

$\epsilon(\omega) = \epsilon_1(\omega) + \epsilon_2(\omega)$



- *Used in waveguide; optical transmission, information storage, nanophotonic devices; switches; resonant light scatterers; sensors, biosensors; surface-enhanced Raman scattering (SERS)*

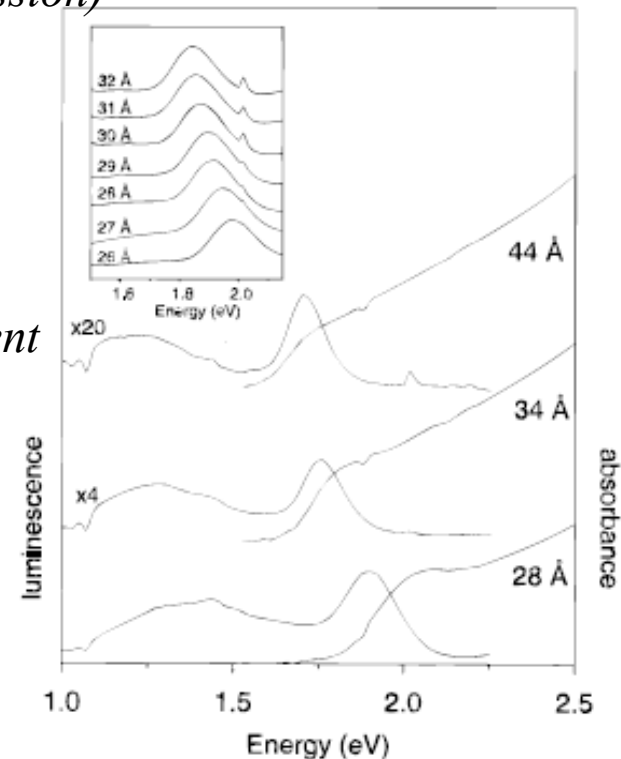
## Optical properties due to size confinement effect

### \* Semiconductors

- Rapidly increase in band gap with a decreasing size
- Optical absorption and photoluminescence (emission)
  - Blue shift

### \* Metals

- Observed in far infra-red absorption measurement for passivated gold nanoparticles with  $<2\text{nm}$  diameter ( $\sim 100$  atoms)



Absorbance and PL spectra for several sizes of InP nanocrystal samples. The PL spectra are composed of a high-energy band edge emission band and a low-energy trapped emission band. The insert shows additional scaled PL spectra for a sequence of sample sizes exhibiting a smooth blue shift of the band edge emission feature with decreasing nanocrystal size. The samples have been treated with decylamine and were exposed to air.

### 3.8 Electrical conductivity

$$\rho_{total} = \rho_{thermal} + \rho_{defect}$$

*Thermal: electron collision with vibrating atoms (phonons),  
temperature dependent*

*Defect: impurities, lattice defect, grain boundary*

#### \* *Surface scattering*

*if the mean free path of electrons,  $\lambda_0 >$  particle size,  $d_p$*

*Elastic (specula) scattering*

*Inelastic (nonspecula, diffuse) scattering  $\rightarrow$  increase in  $\rho$  (resistivity)*

#### \* *Change of electronic structure*

*Quantum size effect  $\rightarrow$  widening and discrete band gap*

*metals  $\rightarrow$  semiconductors, semiconductors  $\rightarrow$  insulators*

*Bi nanowires:  $\sim 52\text{nm}$ ; Si nanowires:  $\sim 15\text{nm}$*



\* *Quantum transport*

Ballistic conduction: transport without any collision

$\lambda_0 >$  length of conductor

Resistance of a carbon nanotube (CNT) can be independent of the tube length.

Coulomb blockade:

Ohm's law  $I = U / R$

$E, U$ : Energy (voltage) required to add a single charge to the particle

$$E = \frac{e^2}{2C(d_p)} \quad U = \frac{e}{C(d_p)} \quad \text{and} \quad I = \frac{U}{R} = \frac{e}{RC(d_p)}$$

where  $C(d_p) = 2\pi d_p \epsilon_0 \epsilon_r$

For bulk materials ( $d_p \sim \infty$ ),  $E \sim 0$  and  $V \sim 0$

“ $I$  “ varies linearly and continuously with”  $R$ ”

Ampere: 1 coulomb/s      cf.  $e = 1.602 \times 10^{-19}$  coulomb

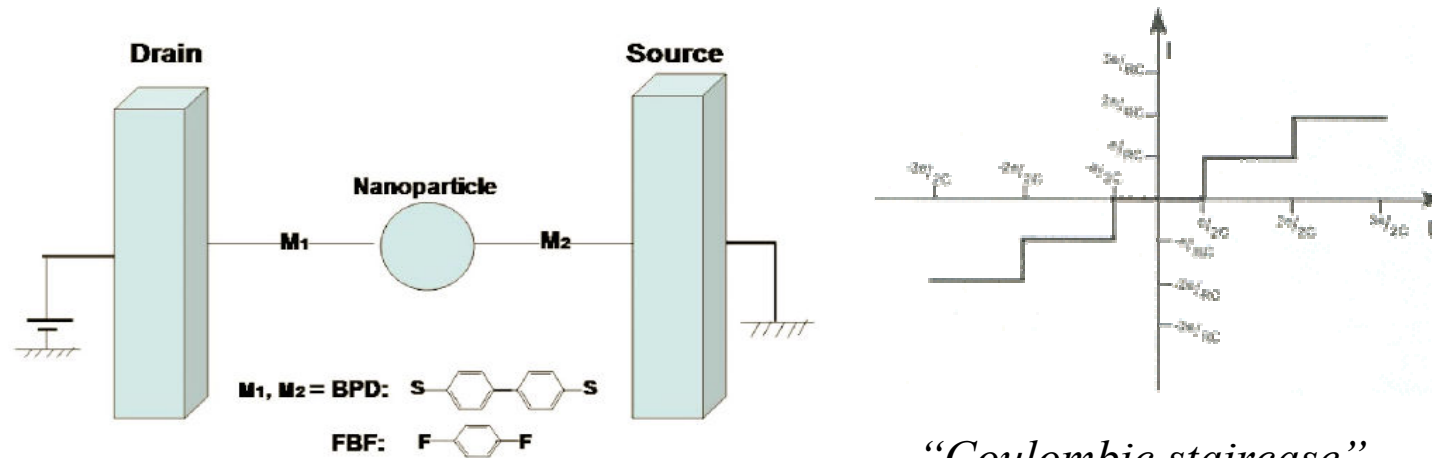
As size decreases,  $E \sim$  finite

A single electron can be added to the particle when  $E \gg kT \dots$

“Coulomb blockade”

Two electrons ( $I = \frac{2e}{RC}$ ) can be added under  $U = \frac{2e}{C}$

For the configuration



“Coulomb staircase”

Tunneling conduction: charge transport through an insulating medium separating two conductors that are extremely closely spaced

- Overlapping of electron wave functions from two conductors

\* Single electron transistor

[http://www.glue.umd.edu/~bekane/QC/QC@UMD's\\_LPS\\_Single%20Electron%20Transistor.htm](http://www.glue.umd.edu/~bekane/QC/QC@UMD's_LPS_Single%20Electron%20Transistor.htm)

### 3.9 Ferromagnetic properties

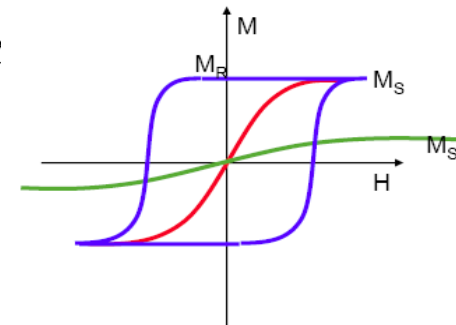
#### \* Ferromagnetic materials

- Unpaired electrons in d-orbit : directed spin  
e.g., Fe, Co, Ni
- Exchange energy: alignment of atomic spins
- Domain theory
  - Lowering magnetostatic energy by forming multiple domains
- Hysteresis loop

$H$ : magnetic field;  $M$ : magnetization (magnetic moment / volume)

Coercivity, saturation magnetization,  $BH$  product

- Magnetic alignment
  - Rotation of magnetic dipoles
  - Movement of domain boundary
- Used for permanent magnet, core transformer



- Ferromagnetisme
- Paramagnetisme
- Superparamagnetism

\* *Single- domain particles*

- *Energy lowering by domain formation is balanced ( $E_{ms} \propto d^3$   $d$ : domain size) by energy increase for boundary formation ( $E_{bf} \propto d^2$ ) as  $d$  decreases...*
- *For small particles, single domain is in the lowest energy state...*

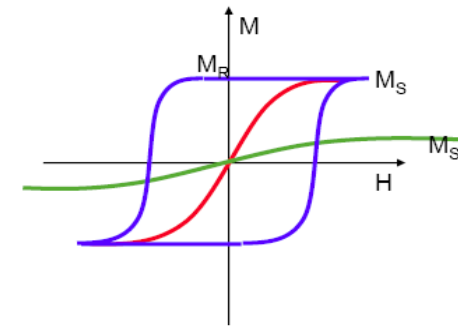
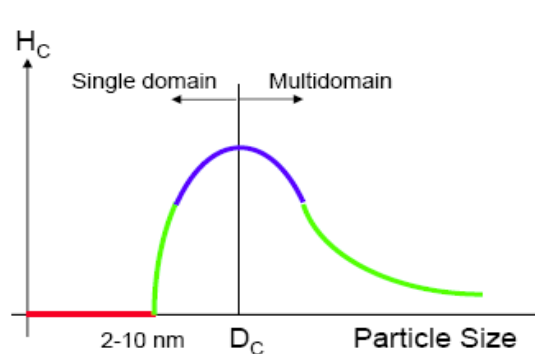
$$d_p : 10 \sim 100\text{nm}$$

- *Used for magnetic recording media*

\* *Superparamagnetism:*

- *Thermal fluctuation > magnetic alignment as the size decreases*

Usually,  $d_p \lesssim 15\text{nm}$



- *No hysteresis loop*

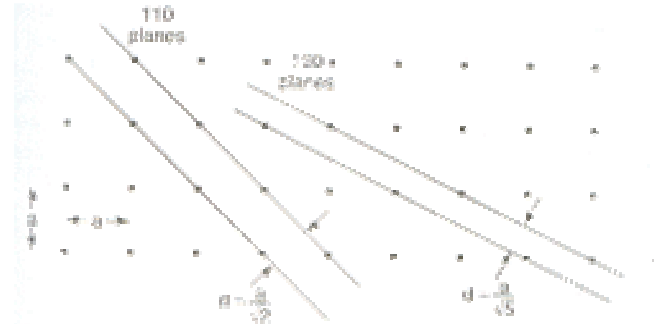
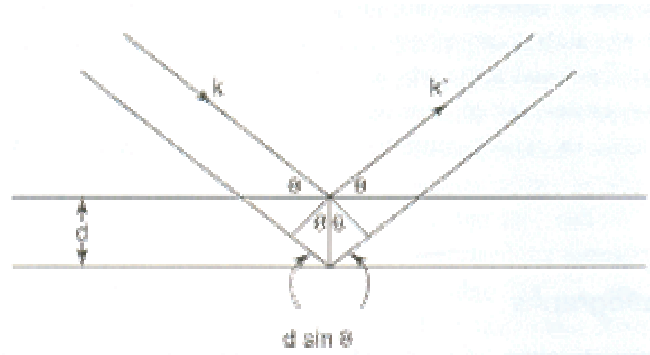
- *Used in biomedical application, ferrofluids, sensors*

- Ferromagnetisme
- Paramagnetisme
- Superparamagnetisme

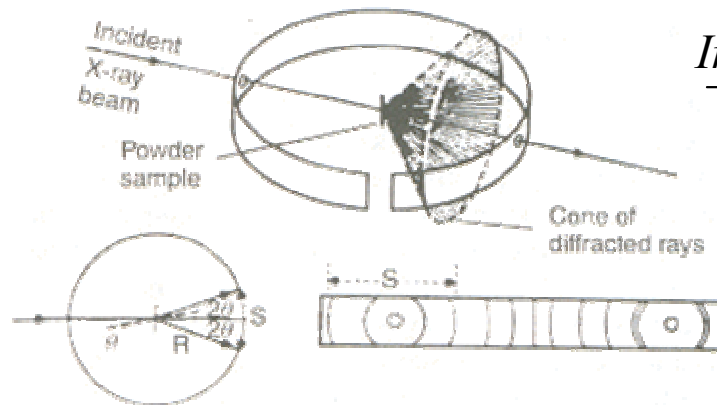
### 3.10 Characterization of nanoparticles

#### (1) X-ray diffraction

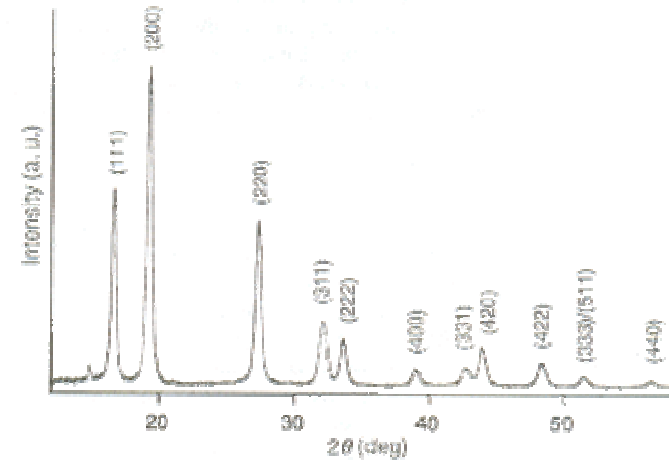
- Bragg's law;  $2d \sin \theta = n\lambda$



- used to identify the specimen's crystalline phases and to measure structural properties



Intensity vs.  $2\theta$



XRD scan of nanocrystalline TiN

- Peak broadening by inhomogeneous strains and crystallite size,  $d_c$

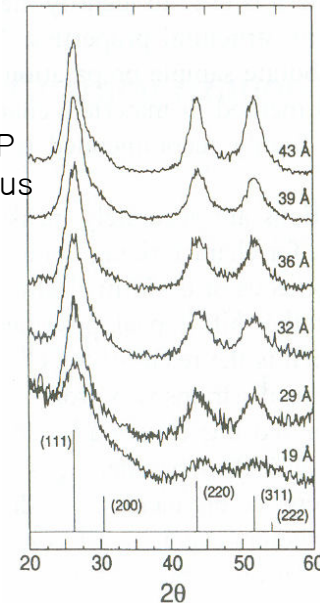
If no inhomogeneous strain,

$$d_c = \frac{K\lambda}{B \cos \theta_B} \quad \text{Scherrer equation}$$

where  $K$ : Scherrer's constant ( $\sim 1$ );  $\lambda$ : X-ray wavelength;

$B$ : half width of the peak;  $\theta_B$ : diffraction angle

XRD patterns of InP particles with various sizes



- Sometimes  $d_c$  is smaller than true  $d_p$  due to twinned structure of the nanoparticles
- yields low intensity, suitable for high  $Z$  materials, requires large amount of materials, gives average values

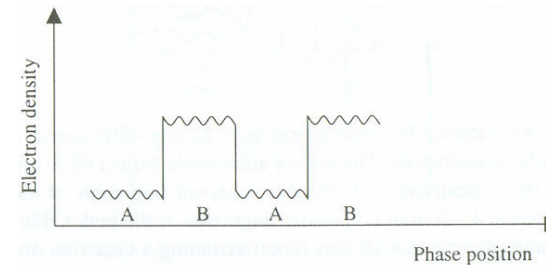
cf. Intensity for electron diffraction intensity  $10^8$  larger than for XRD

## (2) Small angle X-ray scattering

- X-ray intensities from fluctuation in electron density\* over the length  $< 10\text{nm}$ :

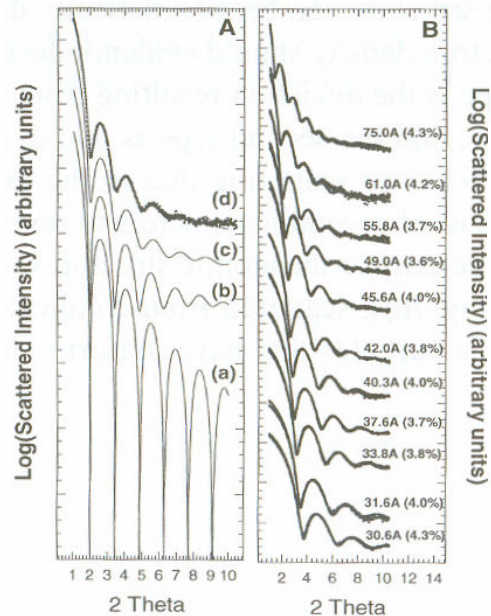
$$2\theta < 5^\circ$$

\* Difference in density and composition

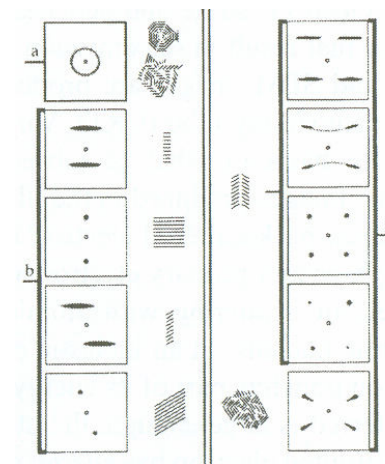


- gives the size and shapes of small particles, their surface area per unit volume, irrespective crystalline or amorphous

- also used for determining size and ordering of mesoporous materials



(A) SAXS patterns for model structures having 4500 atoms, comparable to a 62 Å diameter CdSe nanocrystals. The curves are models for (a) 62 Å spheres of uniform electron density, (b) monodisperse, 4500 atom spherical fragments of the bulk CdSe lattice, (c) monodisperse, 4500 atom ellipsoidal fragments of the bulk CdSe lattice, having a 1.2 aspect ratio, and (d) fit to SAXS data (dots) assuming a Gaussian distribution of ellipsoids (as in curve c), yielding the nanocrystal sample size and size distribution. (B) SAXS patterns for CdSe nanocrystal samples ranging from 30 to 75 Å in diameter (dots). Fits are used to devise the nanocrystal sample size, reported in equivalent diameters, and size distributions, ranging from 3.5 to 4.5% for the samples shown. [C.B. Murray, C.R. Kagan, and M.G. Bawendi, *Ann. Rev. Mater. Sci.* **30**, 545 (2000).]



Structures and SAXS patterns

### (3) Gas adsorption

- Adsorption: reducing surface energy of the adsorbents: physical\* vs. chemical

#### Determination of specific surface area(m<sup>2</sup>/g)

BET adsorption isotherm

$$\frac{1}{v} \frac{x}{1-x} = \frac{c-1}{cv_m} x + \frac{1}{cv_m}$$

where  $v$  = volume of gas adsorbed per gram of sample;  $v_m$  = volume of gas adsorbed at monolayer coverage per gram of sample at standard condition;  $x = P/P_o$

Adsorption of nitrogen at 77K(liquid nitrogen) → A plot of  $(1/v)(x/(1-x))$  versus  $x$

→ from slope and interception →  $v_m$

Then, specific surface area(m<sup>2</sup>/g)

$$S_g (m^2 / g) = \frac{v_m N_0 A_a}{V} = 4.35 v_m (cm^3 / g) \quad \text{for nitrogen at } -196^\circ C$$

where  $N_0$ : Avogadro number;

$A_a$ : area on which nitrogen molecule adsorbed (=0.162nm<sup>2</sup>)

\* Single Point BET Method

Use of  $v_m$  at  $P/P_o = 0.30$  neglecting the intercept of the BET plot (small enough)



Pore volume in porous materials

\* Mercury penetration

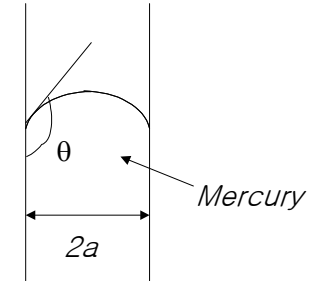
For Pore size 20-1000nm

From force balance

$$\pi a^2 p = -2\pi a \sigma \cos \theta$$

$$\therefore d(\text{nm}) = -\frac{\sigma \cos \theta}{p} = \frac{4.36 \times 10^4}{p(\text{psi})}$$

for mercury

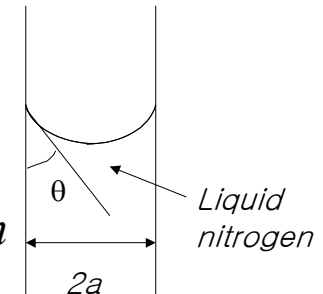


Measurement of mercury penetration volume vs.  $p$  gives pore volume vs.  $a$

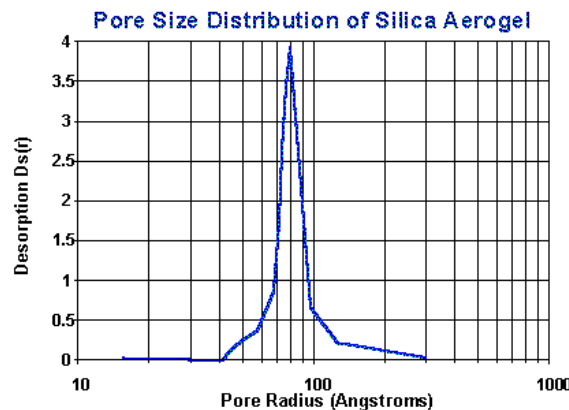
\* Nitrogen adsorption-desorption

Pore size up to 20nm

Kelvin equation  $d(\text{nm}) = \frac{-4\sigma V \cos \theta}{RT \ln(p/p_0)} = 1.904 \left( \log \frac{p_0}{p} \right)^{-1}$  for nitrogen



\* Measurement of desorbed nitrogen volume  $V$  vs.  $p$  gives pore volume vs.  $a$



### (3) Optical spectroscopy

- Absorption, emission, vibrational spectroscopy

Transition from one energy level to the another level in the electronic structure of atoms, molecules and solids

Sharpness of the peaks: atoms > molecules > solids (energy gap between CB and VB)

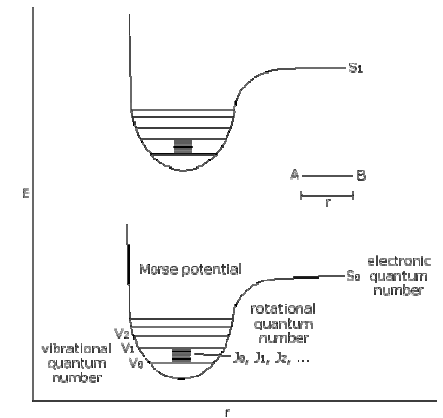
- Transition between electronic levels: UV/VIS and photoluminescence (PL)

\* Infra-red spectroscopy:

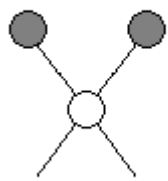
- Transition between vibrational levels ( $10^{12}$  -  $10^{14}$  Hz, or 3-300  $\mu\text{m}$ )

- Caused by change in dipole moment of the molecule

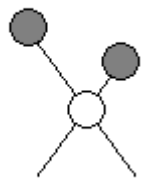
- Stretching and bending



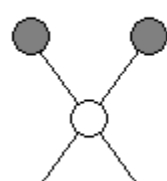
#### Stretching vibrations



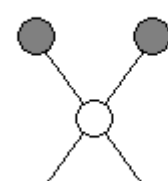
Symmetric



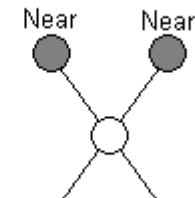
Asymmetric



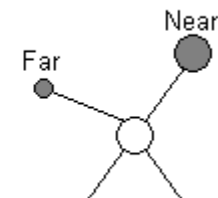
In-plane rocking



In-plane scissoring



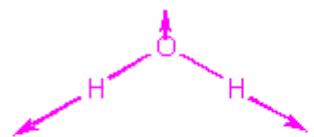
Out-of-plane wagging



Out-of-plane twisting

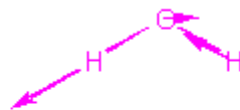
#### Bending vibrations

## Examples



symmetric stretch

$\nu_s$



asymmetric stretch

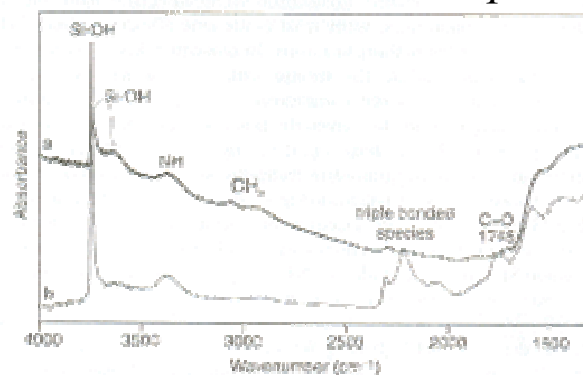
$\nu_{as}$



bend

$\delta$

- FT-IR: Fourier transform of intensity-time output to intensity-frequency data
- Determines the identities, surrounding environments or atomic arrangement, and concentrations of the chemical bond in the sample



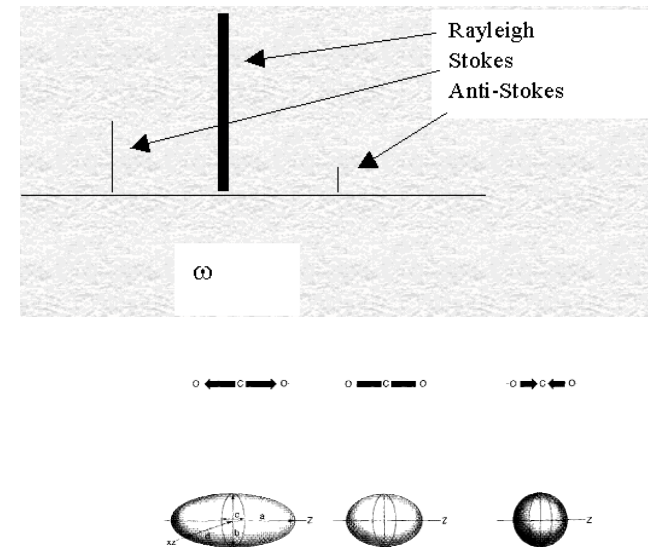
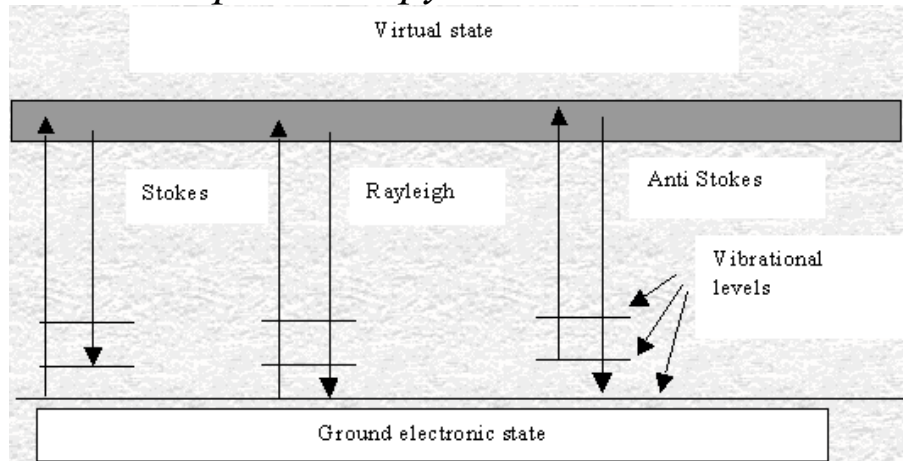
FT-IR spectra of silica particles before and after heat treatment

<http://www.shu.ac.uk/schools/sci/chem/tutorials/molspec/irspec1.htm>

<http://www.le.ac.uk/chemistry/schools/TeachersHO.pdf>

<http://www.wpi.edu/Academics/Depts/Chemistry/Courses/CH2670/infrared.html>

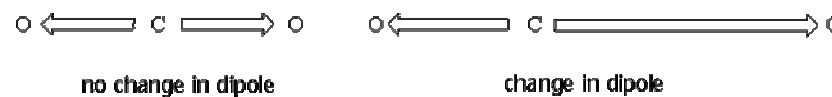
\* *Raman spectroscopy*



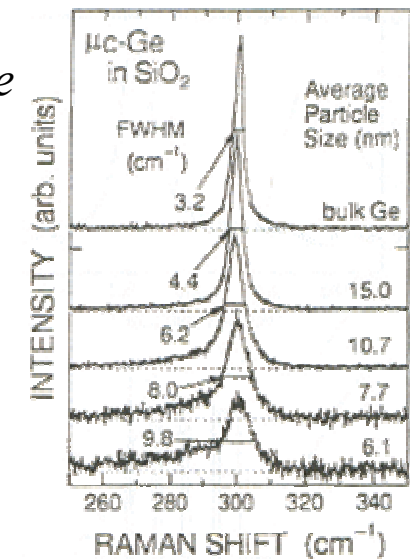
- *Change in polarizability of the bond (electronic mobility)*
- *Used for structural characterization (sensitive to the lengths, strengths and arrangement of chemical bonds, but less sensitive to the chemical composition)*

<http://carbon.cudenver.edu/public/chemistry/classes/chem4538/raman.htm>

*cf. In IR,*



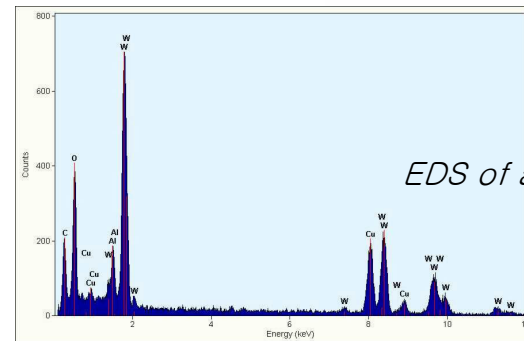
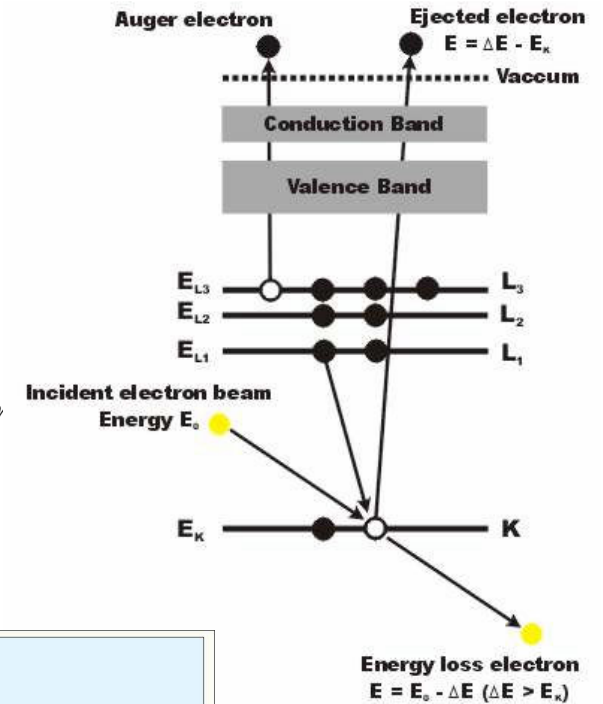
*No signal in IR*



(4) *Electron spectroscopy*

\* *Energy dispersive spectroscopy (EDS) and Auger electron spectroscopy (AES)*

- *Irradiation of X ray or gamma ray ejected an electron from an inner shell*
- *Measuring the energies of the X rays and Auger electrons*



\* *X-ray photoelectron spectroscopy*

- *Also known as Electron Spectroscopy for Chemical Analysis (ESCA)*
- *Photoelectric effect by low energy X ray*
- *Energy of the ejected electron  $E_E = h\nu - E_B$*

*where  $E_B$ : Energy of the bound electron state*